

Effects of silicocalcium on microstructure and properties of Mg-6Al-0.5Mn alloy

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Abstract: The effects of silicocalcium on the microstructure and mechanical properties of casting magnesium alloy Mg-6Al-0.5Mn(AM60) were studied. The results show that the microstructure of AM60 casting magnesium alloy is effectively refined by adding small amount of silicocalcium. The grain size decreases from 180 μm to 80 μm with 1.8% Si addition, while the size increases with 2.5% Si addition. The AM60+Si-Ca alloys mainly contain Mg matrix, $\beta\text{-Mg}_{17}\text{Al}_{12}$ phase, Al_8Mn_5 phase and a small polygonal type Mg_2Si phase in matrix. Al_8Mn_5 can act as the heterogeneous nucleation for the Mg_2Si phase. With the increase of silicocalcium, the content of Mg_2Si phase increases gradually, the Mg_2Si particles grow up and change coarse gradually. The microhardness of AM60 matrix increases with silicocalcium addition. The peak values of the tensile strength, elongation and impact toughness appear simultaneously with 1.8% silicocalcium addition, and the tensile strength, elongation and impact toughness are heightened respectively by 13.9%, 28.5% and 100%.

Key words: silicocalcium; magnesium alloy; AM60 alloy; microstructure; mechanical properties

1 Introduction

Magnesium alloys are attractive for applications in the automobile, aerospace and electronic industries due to their light mass, high stiffness, high specific strength, good dimensional stability and damping capacity. It is the lightest space structural metal[1–5]. The alloy AZ91 is one of the most widely applied materials of the commercial magnesium alloys because it offers a good combination of mechanical and corrosion properties as well as excellent castability. However, this alloy is not suitable for use at high toughness demanded owing to its poor toughness. Although the alloy AM60 has a good toughness and plasticity, its poor strength limits its applications. Therefore, it is pressing to develop some new cast magnesium alloys with high strength, good toughness and low cost.

The Si addition to magnesium alloys causes an increased fluidity of the molten metal. The Mg_2Si formed by the addition of Si exhibits high melting point, high hardness, low density, high elastic modulus and low thermal expansion coefficient[6, 7]. But the existence of

the coarse Chinese script type Mg_2Si phase can lead to the reduction of the mechanical properties of the alloys [8]. A proper addition Ca to magnesium alloys can be efficient in refining the microstructure, and causes a morphological change in Mg_2Si particles from coarse Chinese script shape to a small polygonal type[7, 9, 10]. The related investigations have revealed that the modification mechanism of Ca is polygonal type Mg_2Si particles nucleating from CaSi_2 particles[7]. To import Si and Ca, many of the studies have been realized by Al-Si, Al-Ca or Mg-Ca master alloys[9–11]. However, less work has been carried out on importing Si and Ca by silicocalcium which is applied widely in steel-making.

The present study is to investigate the influence of small addition of silicocalcium on the microstructure and properties of Mg-6Al-0.5Mn-based alloys. The purpose of this study is to produce a high properties and low cost magnesium alloy by a simplified casting process.

2 Experimental

The alloys with composition listed in Table 1 were prepared in a steel crucible with an electric resistance

Table 1 Composition of alloys (mass fraction, %)

Alloy	Nominal					
	Al	Mn	Zn	Si	Ca	Mg
1	6	0.5	0.2	—	—	Bal.
2	6	0.5	0.2	1.0	0.18	Bal.
3	6	0.5	0.2	1.8	0.32	Bal.
4	6	0.5	0.2	2.5	0.45	Bal.

Alloy	Analyzed					
	Al	Mn	Zn	Si	Ca	Mg
1	6.36	0.32	0.25	—	—	Bal.
2	6.48	0.28	0.22	0.91	0.08	Bal.
3	6.53	0.34	0.23	1.62	0.17	Bal.
4	6.40	0.32	0.24	2.38	0.26	Bal.

furnace protected by $\text{CO}_2\text{-}0.5\%\text{SF}_6$ from commercially pure magnesium, aluminum, zinc, and silicocalcium. Manganese was added as Al-10Mn master alloy. The melt was held at 720 °C, and silicocalcium was added at 780 °C, then poured into 30 mm-diameter steel moulds at 720 °C. The composition of silicocalcium alloys is listed in Table 2.

Table 2 Composition of Si-Ca alloys(mass fraction, %)*

Ca	Si	C	Al	P	S
30	55–65	< 0.8	< 2.4	< 0.04	< 0.06

* National Standard YB/T5051 - 1997

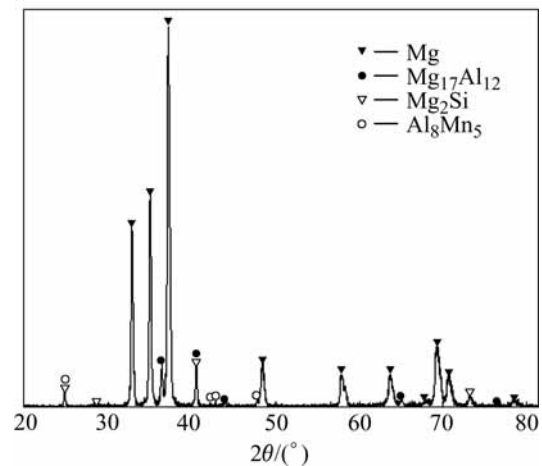
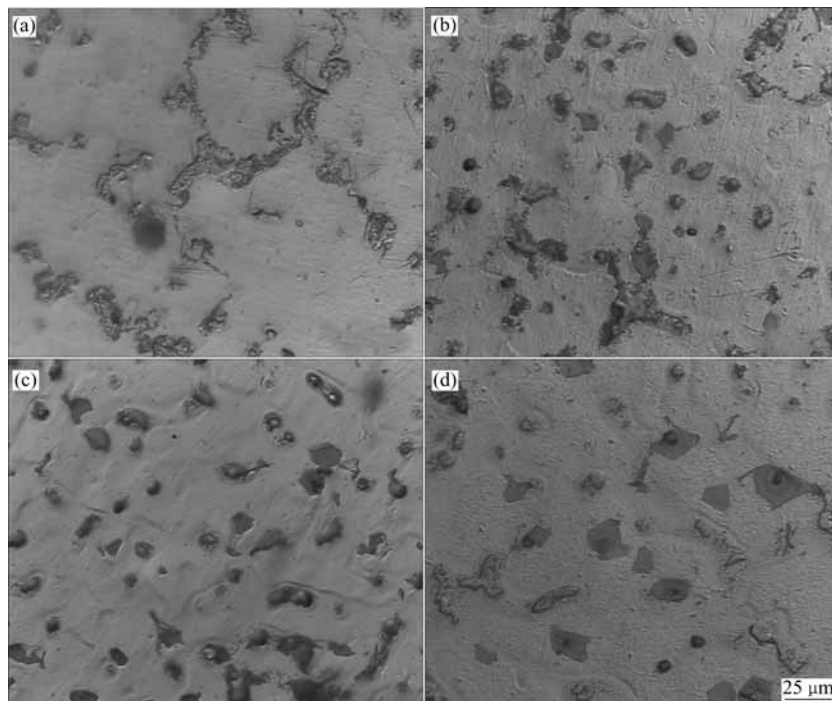
In this work the homogenization heat treatment(T4) for as-cast specimens was performed at 410 °C for 12 h followed by water quench. The microstructural analysis was carried out using an optical microscopy (Olympus)

and a scanning electron microscope (JSM-5610LV) equipped with an energy dispersive X-ray spectrometer (EDS). Phases were analyzed by Philips PW1700 type XRD analyzer operated at 40 kV and 40 mA. The microhardness test was carried out by HV1000 type microhardness instrument with 0.98 N loading.

3 Results and discussion

3.1 Microstructure characteristic

The XRD spectrum (Fig.1) of AM60+2.5Si-Ca alloy shows that the main phases are α -Mg phase, β - $\text{Mg}_{17}\text{Al}_{12}$ phase, Al_8Mn_5 phase and Mg_2Si phase. The Ca of silicocalcium does not result in the formation of any new phase. Fig.2(a) shows the microstructure of

**Fig.1** XRD pattern of as-cast AM60+2.5Si-Ca alloy**Fig.2** Microstructure of as-cast alloys: (a) AM60; (b) AM60+1.0Si-Ca; (c) AM60+1.8Si-Ca; (d) AM60+2.5Si-Ca

permanent mould cast alloy 1. It can be seen that its microstructure consists of ashen α -Mg matrix, discontinuous β - $Mg_{17}Al_{12}$ phase which distributes along the grain boundaries, and a small quantity of Al-Mn particle phase. With the small addition of silicocalcium (Figs.2(b), (c), (d)), many small polygonal type phases are found in matrix and grain boundaries. Fig.3 shows the SEM image of alloy 4, and the phases identified in the as-cast samples by EDS are listed in Table 3. It can be seen that the small polygonal type phase is enriched in Mg and Si, and the Mg/Si molar ratio in this phase is nearly 2/1, so this phase is Mg_2Si phase. The result shows that polygonal type Mg_2Si phase can be gained directly with silicocalcium addition. The coarse Chinese script type Mg_2Si phase like literature reported is not found[8]. In addition, many white particle phase can be found in matrix from SEM image. According to the XRD spectrum and EDS results, these phases are Al_8Mn_5 phase.

3.2 Effect of silicocalcium addition on microstructure

Fig.2 shows the change of as-cast microstructure with different silicocalcium addition. When the content of Si is less than 1.8%, with the increase of silicocalcium, the fraction of $Mg_{17}Al_{12}$ phase increases gradually, the $Mg_{17}Al_{12}$ phase becomes finer and its distribution is

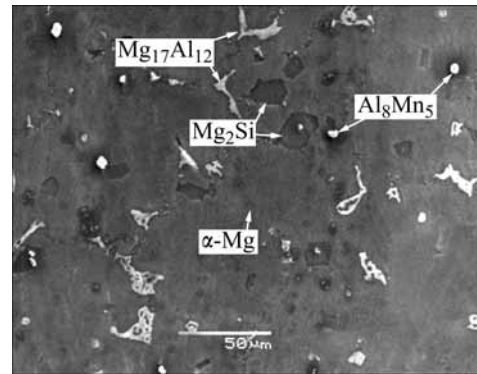


Fig.3 SEM Micrograph of AM60+2.5Si-Ca alloy

Table 3 Chemical compositions of phases in Fig.3 by EDS (molar fraction, %)

Phase	Mg	Al	Si	Ca	Mn
$Mg_{17}Al_{12}$	67.58	31.56	0	0	0
Mg_2Si	66.37	0	33.63	0	0
Al_8Mn_5	11.80	41.22	5.36	0.6	37.93
α -Mg	92.11	6.37	1.53	0	0

more dispersive, and the grain size decreases gradually from 180 μm to 80 μm (Fig.4). This may be explained by that a large number of fine Mg_2Si particles with high melting point start to solidify and precipitate firstly in the

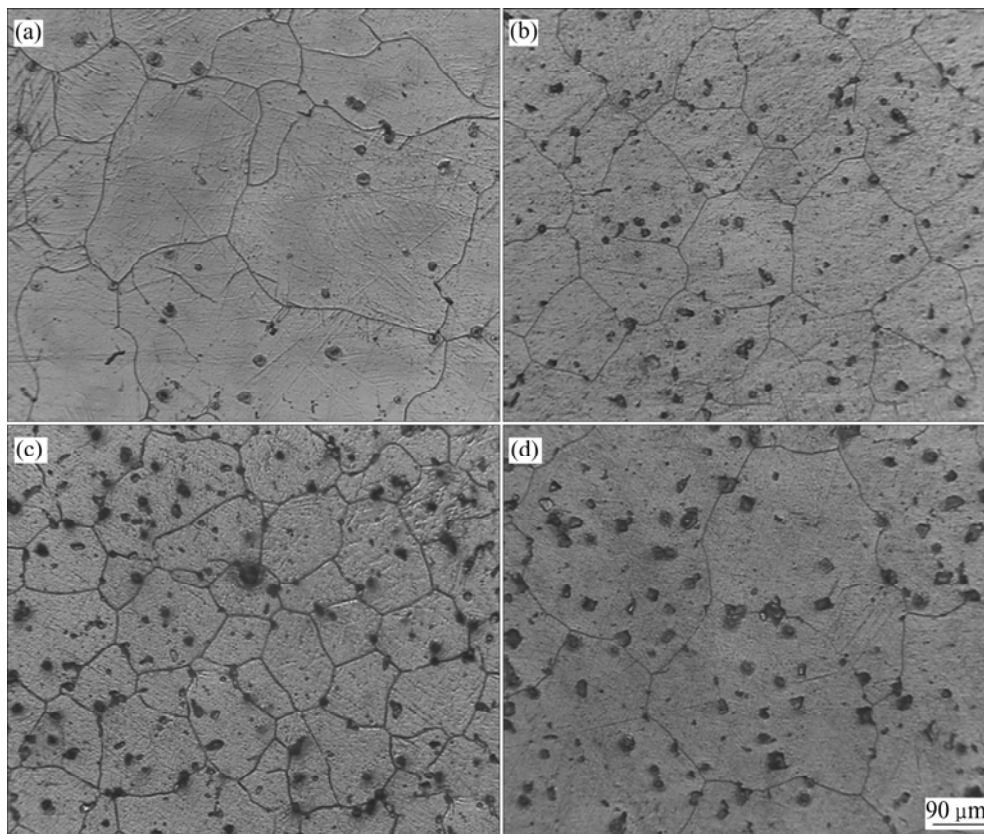


Fig.4 Microstructures of solution-treated alloys at 410 $^{\circ}C$ for 12 h: (a) AM60; (b) AM60+1.0Si-Ca; (c) AM60+1.8Si-Ca; (d) AM60+2.5Si-Ca

process of solidification. The Mg_2Si particles enrich near the grain boundary and pin the grain growing. At the same time, Mg_2Si consumes the Mg atoms of grain boundary, which leads to the increasing of Al atoms relatively and the increasing of the content of $Mg_{17}Al_{12}$ phase finally. On the other hand, Ca mainly dissolves into the $Mg_{17}Al_{12}$ phase, which effectively suppresses the discontinuous precipitation of $Mg_{17}Al_{12}$ in alloys[12,13]. When the content of Si is 2.5%, the distribution of $Mg_{17}Al_{12}$ phase is anymore dispersive but discontinuity reticulation because of its multiplication. With the increase of silicocalcium, the content of Mg_2Si phase increases gradually and its morphology changes coarse.

3.3 Heterogeneous nucleation for Mg_2Si phase

It can be seen that Mg_2Si particles contain small particles inside, which presumably act as nucleation sites for Mg_2Si particle (Fig.5). The EDS analysis shows that this phase consists of Al, Mn, Mg and Si, does not contain Ca. So it is not $CaSi_2$ [7]. According to the Al/Mn molar ratio, we think the nucleus is Al_8Mn_5 phase. That Al_8Mn_5 can act as the heterogeneous nucleation for the Mg_2Si phase has not been reported.

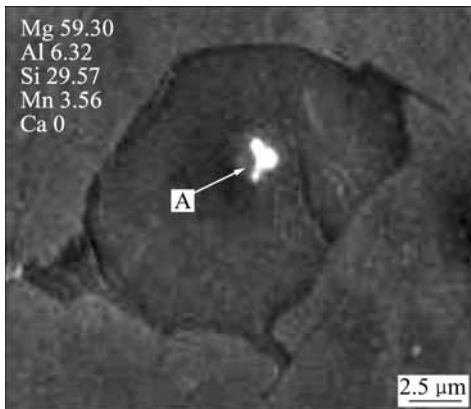


Fig.5 SEM morphology and EDS results of Mg_2Si particle in AM60+2.5Si-Ca alloy

According to the work by BRAMFITT[14], the mathematical model of the two-dimensional lattice misfit is

$$\delta_{(hkl)_n}^{(hkl)_s} = \frac{1}{3} \sum_{i=1}^3 \frac{|d[uvw]_s^i \cos \theta - d[uvw]_n^i|}{d[uvw]_n^i} \times 100\% \quad (1)$$

where $(hkl)_s$ is the low index plane of the substrate, $[uvw]_s$ is the low index direction in $(hkl)_s$; $(hkl)_n$ is the low index plane in the nucleated solid, $[uvw]_n$ is the low index direction in $(hkl)_n$; $d[uvw]_s$ and $d[uvw]_n$ are the atomic spacing along the $[uvw]_s$ and the $[uvw]_n$; θ is the angle between the $[uvw]_s$ and $[uvw]_n$. The result of BRAMFITT's research shows that: one criterion of heterogeneous nucleation is that the disregistry of

nucleant planes is less than 15%[14].

Al_8Mn_5 phase is cubic crystal structure, $a=0.9012$ nm; Mg_2Si phase is face cubic crystal structure, $a=0.6347$ nm. The result of calculation of planar disregistry δ is 41.89% when the orientations relationship between Mg_2Si phase and Al_8Mn_5 phase is $(001)_{Al_8Mn_5} // (001)_{Mg_2Si}$, which is more than 15%. But when the crystal lattice is circumrotated 45° [15](Fig.6), in other word, when the orientations relationship is $[100]_{Al_8Mn_5} // [110]_{Mg_2Si}$, the disregistry is only 0.4%, which is less than 6%. Therefore, Al_8Mn_5 can act as the heterogeneous nucleation for the Mg_2Si phase by this orientation relationship.

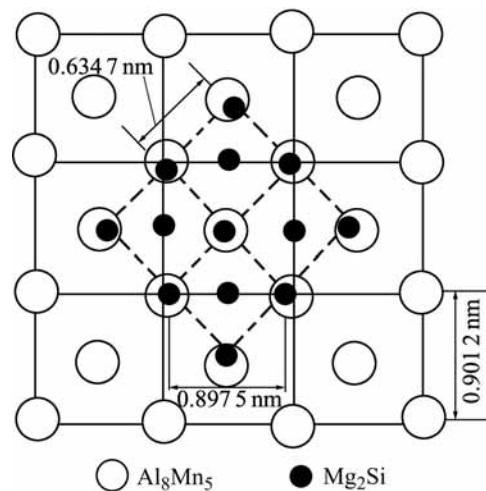


Fig.6 Relation of crystal structure between Al_8Mn_5 and Mg_2Si

Therefore, the morphological change in Mg_2Si particles from coarse Chinese script shape to a small polygonal type can be explained in two aspects: 1) the modification of calcium in silicocalcium for Mg_2Si in the solidification process[7], 2) the modification that Al_8Mn_5 particles act as the heterogeneous nucleation for the Mg_2Si phase.

3.4 Mechanical properties

The microhardness of AM60+xSi-Ca alloys is shown in Fig.7. It can be seen that the silicocalcium addition to the alloy results in the improving of the microhardness of the AM60 alloy. The improvement of microhardness is mainly ascribed to two aspects: 1) The silicocalcium addition promotes the diffusion of the Al atoms to α -Mg matrix; 2) The Mg_2Si phase with dispersive distribution strengthens the matrix.

The tensile properties and impacts toughness of AM60+xSi-Ca alloys are shown in Fig.8. It can be seen from Fig.8 that the tensile strength, elongation and impacts toughness of the AM60 alloys are significantly improved with silicocalcium addition. The peak values of the tensile strength, elongation and impacts toughness appear simultaneously with 1.8% silicocalcium addition,

and the tensile strength, elongation and impact toughness are heightened respectively by 13.9%, 28.5% and 100%, which accords with the grain size forenamed. It can be mainly ascribed to the fine-grain strengthening due to grain-boundaries blocking dislocations and the dispersive Mg_2Si particles due to strengthening matrix.

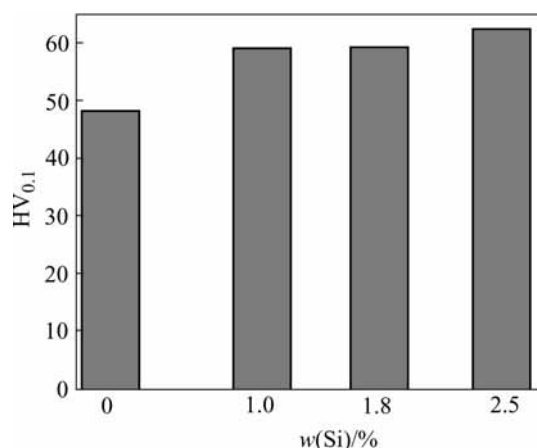


Fig.7 Microhardness of as-cast AM60+xSi-Ca alloys

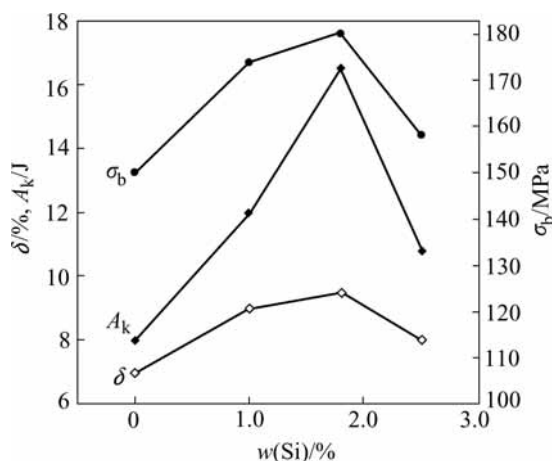


Fig.8 Mechanical properties of as-cast AM60+xSi-Ca alloys

4 Conclusions

1) The addition of silicocalcium is found to be efficient in refining the microstructure of Mg-6Al-0.5Mn alloys. The $Mg_{17}Al_{12}$ phase becomes fine and its distribution becomes dispersive. When the content of Si is less than 1.8%, the grain size decrease from 180 μm to 80 μm .

2) The small polygonal type Mg_2Si can be gained directly with the addition of silicocalcium, which causes the modification of calcium in the silicocalcium in solidification process and Al_8Mn_5 particles to act as the heterogeneous nucleation for the Mg_2Si phase.

3) The microhardness of the Mg-6Al-0.5Mn is improved with the addition of silicocalcium, and the microhardness of the Mg-6Al-0.5Mn with the addition of 2.5% is increased by 27%.

4) The peak values of the tensile strength, elongation and impact toughness appear simultaneously with 1.8% silicocalcium addition, and the tensile strength, elongation and impacts toughness are heightened respectively by 13.9%, 28.5% and 100%.

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