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Preparation of $Al_{2-x}Y_{x}W_{3}O_{12}$ powders by citrate sol-gel process

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Abstract: $Al_{2-x}Y_xW_3O_{12}$ (*x*=0.2, 0.5, 0.8, 1.0, 1.2, 1.5, 1.7 and 2.0) powders were synthesized by citrate sol-gel process. The concentration of species in a citric solution for preparing $Al_{2-x}Y_xW_3O_{12}$ powders was calculated. The powders were characterized by differential thermal analysis(DTA), thermogravimetry(TG), X-ray diffractometry(XRD) and scanning electron microscopy(SEM), respectively. No solid solution of $Al_{2-x}Y_xW_3O_{12}$ is formed with *x* values varying from 0 to 2.0. The maximum solid solubility of Y_2O_3 in $Al_2W_3O_{12}$ and Al_2O_3 in $Y_2W_3O_{12}$ is less than 0.5. $Y_2W_3O_{12}$ easily absorbs water in air and forms a composition of $Y_2W_3O_{12}$ ·3.2H₂O, and $Al_2W_3O_{12}$ forms $Al_2W_3O_{12}$ ·0.17H₂O in the same condition.

Key words: $Al_{2-x}Y_{x}W_{3}O_{12}$; citrate sol-gel; preparation; hydration

1 Introduction

Most substances expand during heating. Recently, materials that exhibit negative thermal expansion(NTE) have attracted considerable interest because of the potential to prepare composite materials with a specific coefficient of thermal expansion, positive, negative, or zero in some specific temperature range. There are many materials that exhibit negative thermal expansion such as zeolites, β -quartz, some AMO₃ oxides (such as BaTiO₃, PbTiO₃) with the perovskite structure[1], compounds with general formula A₂(MO₄)₃ (A: trivalent cations and M: W⁶⁺, Mo⁶⁺, etc)[2–3], and compounds such as ZrMo₂O₈, HfMo₂O₈, Zn₂GeO₄[4] and Zr_{1-x}Hf_xW₂O₈[5], ZrW₂O₈[6–7].

It is well known that the Pechini or citrate sol-gel process is usually considered to have the advantage of even mixing of starting materials, and easy control of the accurate composition of the final products[8–11]. To the knowledge of authors, there are still very few reports on the preparation of $Y_2W_3O_{12}$ and $Al_2W_3O_{12}$ powders by a homogeneous sol-gel process, as most of reported $Y_2W_3O_{12}$ and $Al_2W_3O_{12}$ powders are synthesized by classical solid-mixing-reaction method[12–16]. In this work, the citrate sol-gel process is used to prepare $Al_{2-x}Y_xW_3O_{12}$ powders. A model is used to evaluate the influence of the pH value and citric acid content on the

concentration of the cations citrate complexes in the starting solution. And the hydration of prepared $Al_2W_3O_{12}$ and $Y_2W_3O_{12}$ powders in air are also investigated.

2 Experimental

The raw materials utilized were analytical reagent grade yttrium nitrate (Y(NO₃)₃·6H₂O), aluminum nitrate (Al(NO₃)₃·9H₂O), ammonium tungstate ((NH₄)₅H₅-[H₂(WO₄)]₆·H₂O) and citric acid (H₃Cit). Stoichiometric amounts of yttrium nitrate and aluminum nitrate were dissolved in distilled water, and then citric acid was added. After complete mixing, ammonium tungstate and ammonia solution (about 1 mol/L) were added to get a homogenous transparent solution within a few seconds. The solution was slowly evaporated to form a highly viscous colloid, followed by heating in a temperature range of 120–140 °C for 24 h to obtain a dried gel. Finally, the dried gel precursor was calcined at 900 °C for 10 h to obtain Al_{2-x}Y_xW₃O₁₂ powders.

Two series of specimens were prepared for XRD tests: 1) For series A, XRD characterization was immediately carried out as soon as the sintering process was finished in order to avoid the absorption of the water; 2) For series B, XRD tests were carried out after the prepared $Al_{2-x}Y_xW_3O_{12}$ powders were held in air for about 6 months to study the moisture absorption properties

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of $Al_{2-x}Y_xW_3O_{12}$ powders. The samples were characterized by PHILIPS (Model X'Pert PRO) diffractometer using Cu K_a radiation under 40 kV and 40 mA. The XRD data were collected in the 2 θ range from 10° to 70° with a step of 0.016°.

Differential thermal analysis and thermogravimetric analysis of the precursor were carried out on a NETZSCH STA-449C Thermal Analysis System with a heating rate of 10 °C/min in flowing air with sample mass of 30 mg. Platinum crucibles were used in TG-DTA analysis and the reference material was alpha-alumina. The powder morphology was observed with scanning electron microscopy(SEM) (Model JSM-5610LV, JEOL, Japan).

3 Calculation of concentration of yttrium and aluminum citrate complexes in Al³⁺-Y³⁺-H₃Cit solution

For $Al_{2-x}Y_xW_3O_{12}$ preparation, there are several kinds of ions, such as, Y^{3+} , $Y(H_2Cit)^{2+}$, $Y(OH)^{2+}$, $[Al^{3+}]$, Al(Cit), $Al(HCit)^+$, $Al(OHCit)^-$, $Al(OH)^{2+}$, $Al(OH)_{2+}^+$,

H₃Cit, H₂Cit⁻, HCit²⁻ and Cit³⁻ existing in Al³⁺-Y³⁺-H₃Cit solution. In order to achieve the complete homogeneity of yttrium and aluminum ions at atomic level and accurate control of the precursor composition, it is very important to increase the relative concentration of yttrium and aluminum citrate complexes, such as $Y(H_2Cit)^{2+}$, Al(Cit) and Al(OHCit)⁻ and decreases the relative concentration of Al³⁺ and Y³⁺. In the present work, a theoretical model similar to that in Ref.[17] was used to calculate the concentration of yttrium and aluminum citrate complexes in Al³⁺-Y³⁺-H₃Cit solution under different conditions. The concentration of citric acid and pH value of the solution were taken into consideration.

For $Al_{1.0}Y_{1.0}W_3O_{12}$ preparation, the theoretically calculated yttrium and aluminum complex concentrations at different pH values and concentrations of citric acid are shown in Figs.1 and 2. The following phenomena can be seen from these figures.

1) The concentration of Y^{3+} and Al(Cit) ions decreases with pH values increasing till pH=5 at first, then the concentration of Y^{3+} ions increases evidently till

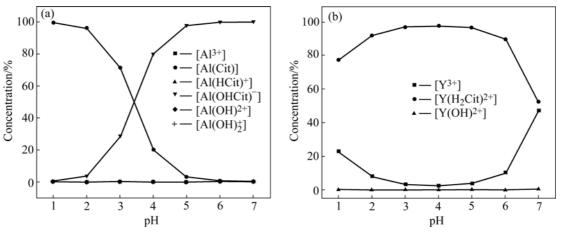


Fig.1 Evaluated concentration of aluminum and yttrium species in solution with 1:1 molar ratio of citric acid to total cations: (a) Al^{3+} -H₃Cit; (b) Y^{3+} -H₃Cit

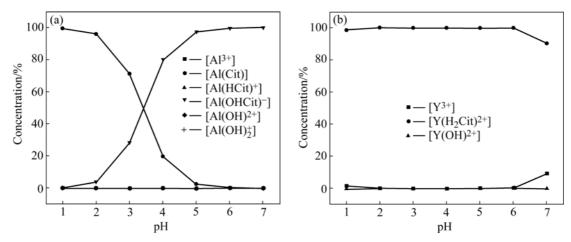


Fig.2 Evaluated concentration of aluminum and yttrium species in solution with 3:1 molar ratio of citric acid to total cations: (a) Al^{3+} -H₃Cit; (b) Y^{3+} -H₃Cit

pH=7. The concentration of Al(Cit) ions is about zero from pH=5 to 7. The concentration of Al(OHCit)⁻ ions increases with pH increasing. The concentration of $Y(H_2Cit)^{2+}$ increases with pH increasing until pH=5, and then decreases at pH=7. The percentage of $[Y(H_2Cit)^{2+}]$ to total yttrium concentration at pH=7 is only about 50% with the 1:1 molar ratio of citric acid to the total cations (Fig.1).

2) The concentration of citric acid shows a little influence on the concentration of Y^{3+} and $Y(H_2Cit)^{2+}$ ions. The concentration of Y^{3+} ion decreases with increasing the citric acid content at the same pH value.

3) The concentrations of $Y(OH)^{2^+}$, Al^{3^+} , $Al(OH)^{2^+}$, $Al(OH)_2^{+^+}$, $Al(HCit)^+$ ions are about lower than zero compared with those of $Y(H_2Cit)^{2^+}$, Al(Cit), $Al(OHCit)^-$ in all conditions.

4) At pH=4 and $[H_3Cit]_T/{[Y]_T+[Al]_T}=3$, almost all yttrium and aluminum ions completely form yttrium citrate complex $Y(H_2Cit)^{2+}$ and aluminum citrate complex Al(OHCit)⁻, Al(Cit).

According to the theoretically calculation (Figs.1 and 2), at $[H_3Cit]_T/{[Y]_T+[Al]_T}=3$ and pH=4, the percentage of the yttrium and aluminum citrate complex would increase to about 100%, indicating that $[H_3Cit]_T/{[Y]_T+[Al]_T}=3:1$ and pH=4 are suitable for $Al_{2-x}Y_x$ - W_3O_{12} precursor synthesis. The calculation results are in agreement with the experimental phenomena, i.e. a transparent sol and gel with no precipitation is easily formed at the pH=4 and $[H_3Cit]_T/{[Y]_T+[Al]_T}=3:1$ in the experiment.

4 Results and discussion

4.1 TG-DTA analyses of precursor

The TG-DTA curves of the dry Al₁₀Y₁₀W₃O₁₂ precursor with pH=4 and $[H_3Cit]_T / \{[Y]_T + [AI]_T\} = 3:1$ are shown in Fig.3. The mass loss can be divided into two temperature regions, namely, from room temperature to 225 °C, and from 225 °C to 751 °C. These two regions correspond to mass loss of about 40%, and 25% respectively. At higher temperature, there is almost no change in mass, which can be attributed to the formation of a pure oxide system. The exothermic peak in the first temperature region (about 225.0 °C) can be contributed to the decomposition of the citrate $Y(H_2Cit)^{2+}$, Al((OH)Cit⁻), Al(Cit), $(NH_4)_5H_5[H_2(WO_4)]_6H_2O$ and the formation of the aluminum and yttrium carbonate. The broad exothermic peaks in the second temperature region from about 400 °C to 900 °C may be caused by the decomposition of the carbonate and the formation of Al₂W₃O₁₂ and Y₂W₃O₁₂.

4.2 XRD analyses of Al_{2-x}Y_xW₃O₁₂ powders

The X-ray diffraction patterns of prepared

Al_{2-x}Y_xW₃O₁₂ calcined at 900 °C for 10 h are shown in Fig.4. It can be seen that, 1) pure Al₂W₃O₁₂ and Y₂W₃O₁₂ are prepared by citrate sol-gel methods; and 2) Al_{2-x}Y_xW₃O₁₂ can not form a complete solid solution as *x* values vary continuously from 0.0 to 2.0. The solid solubility limit of Y₂O₃ in Al₂W₃O₁₂ is less than 0.5 (molar fraction), and so is that of Al₂O₃ in Y₂W₃O₁₂.

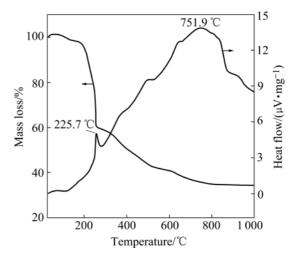


Fig.3 TG-DTA curves of Al_{1.0}Y_{1.0}W₃O₁₂ precursor

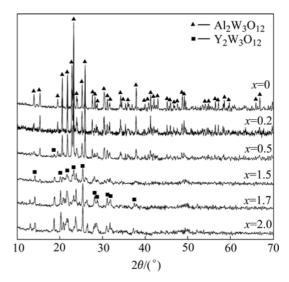


Fig.4 XRD patterns of $Al_{2-x}Y_xW_3O_{12}$ precursors annealed at 900 °C for 10 h

The XRD patterns of series A and series B of $Al_2W_3O_{12}$ and $Y_2W_3O_{12}$ specimens are shown in Fig.5. It is shown that all the diffraction peaks of series B samples (even the sample re-annealed at 800 °C) shift to high degree, indicating that the crystal cell volume of series B samples is less than that of series A samples. This result shows that water molecules can be present in the crystallographic voids and cause shrinkage of the structure. The difference in the volume of the hydrated (series B samples) and the unhydrated (series A samples) phase indicates that the water molecules cause shrinkage

of the framework structure. It is possible that the Y–O–W linkages are bent away from 180° angle in the hydrated phase. As the water is lost, the Y–O–W angle approaches the 180° linkage angle.

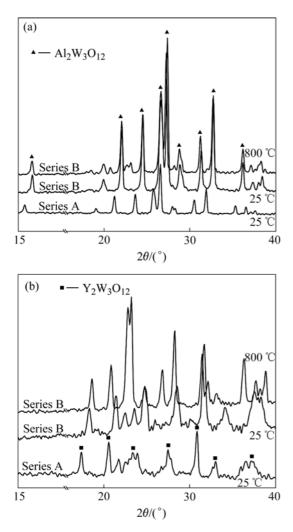


Fig.5 XRD patterns of $Al_2W_3O_{12}$ and $Y_2W_3O_{12}$ under different condition: (a) $Al_2W_3O_{12}$; (b) $Y_2W_3O_{12}$

4.3 Moisture absorbing properties of Al₂W₃O₁₂ and Y₂W₃O₁₂ powders

TG-DTA analysis of B series $Al_2W_3O_{12}$ and $Y_2W_3O_{12}$ samples is shown in Fig.6. Based on calculation, the molecule number of absorbed water in B series $Y_2W_3O_{12}$ sample hold in air for 6 months is about 3.2, which is in good agreement with the results of KARMAKAR et al[13]. The molecule number of absorbed water in B series $Al_2W_3O_{12}$ sample is about 0.17.

4.4 SEM characterization of $Al_2W_3O_{12}$ and $Y_2W_3O_{12}$ powders

The SEM micrographs of $Al_2W_3O_{12}$ and $Y_2W_3O_{12}$ samples annealed at 900 °C for 10 h are shown in Fig.7. The particles of $Al_2W_3O_{12}$ and $Y_2W_3O_{12}$ powders are

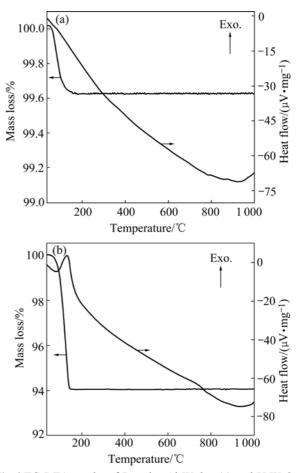


Fig.6 TG-DTA results of B series $Al_2W_3O_{12}$ (a) and $Y_2W_3O_{12}$ (b) samples

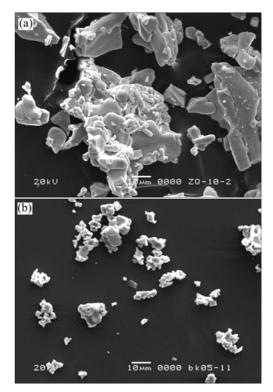


Fig.7 SEM photographs of $Al_2W_3O_{12}$ (a) and $Y_2W_3O_{12}$ (b) powder prepared at 900 $\,^\circ\!C$ for 10 h

widely distributed with the larger ones of more than 40 μ m and the smaller ones of just 3–5 μ m. It is also seen from Fig.7 that the particle size of Al₂W₃O₁₂ is much larger than that of Y₂W₃O₁₂.

5 Conclusions

1) The calculation results show that increasing the concentration of citric acid and pH values can increase the amount of yttrium and aluminum citrate, which plays a positive role in the formation of $Al_{2-x}Y_xW_3O_{12}$ gel. The concentration of those different cation citrate complexes decided by citric acid and pH values in the starting solution influences the formation of sol and further affects precipitation and segregation during gelling and charring.

2) $Al_{2-x}Y_xW_3O_{12}$ powders are formed by citrate sol-gel process using yttrium nitrate, aluminum nitrate, ammonium tungstate and citric acid as starting materials. The optimum conditions for $Al_{2-x}Y_xW_3O_{12}$ powder synthesis are $[H_3Cit]_T / \{[Y]_T + [Al]_T\} = 3:1$, pH=4 and calcining temperature of 900 °C.

3) The maximum solid solubility of Y_2O_3 in $Al_2W_3O_{12}$ and Al_2O_3 in $Y_2W_3O_{12}$ is less than 0.5. $Y_2W_3O_{12}$ powders absorb much more water than $Al_2W_3O_{12}$ powders in air under the same condition. $Y_2W_3O_{12}$ ·3.2H₂O and $Al_2W_3O_{12}$ ·0.17H₂O form from $Y_2W_3O_{12}$ and $Al_2W_3O_{12}$ respectively after being left in air for 6 months.

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