

Low-temperature performance and high-rate discharge capability of AB₅-type non-stoichiometric hydrogen storage alloy

LU Yan-jing(陆延静), ZHU Lei(朱磊), CHENG Yan(成艳),
CHEN Hui(陈晖), JIAN Xu-yu(简旭宇), WANG Zhong(王忠)

Energy Material and Technology Research Institute, General Research Institute for Nonferrous Metals,
Beijing 100088, China

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Abstract: Low-temperature performance and high-rate discharge capability of AB₅-type non-stoichiometric hydrogen storage are studied. X-ray diffraction(XRD), pressure-composition-temperature(PCT) curves and electrochemical impedance spectroscopy(EIS) are applied to characterize the electrochemical properties of AB_x ($x=4.8, 4.9, 5.0, 5.1, 5.2$) alloys. The results show that the non-stoichiometric alloys exhibit better electrochemical properties compared with that of the AB₅ alloy.

Key words: AB₅-type hydrogen storage alloy; non-stoichiometry; low-temperature performance; high-rate discharge capability

1 Introduction

In recent years, nickel-metal hydride(Ni-MH) secondary batteries have been widely used in various portable electronic devices, electric tools and vehicles because of their excellent electrochemical performance and environmental compatibility[1]. However, the low-temperature performance of Ni-MH batteries is unsatisfied. As we known, hydrogen storage alloys play a key role in the discharging process at low temperature for Ni-MH battery. At present, the commercialized negative electrode materials are mainly AB₅-type alloys due to its long-term cycling stability, high-rate capacity, and good charge-discharge kinetics[2–5]. Therefore, it is necessary to improve the low-temperature electrochemical performances of hydrogen storage alloy.

To our knowledge, no systematically investigation has been done for the AB₅-type non-stoichiometric alloys under high rate and low temperature. In this work, the performance of the AB₅-type non-stoichiometric alloys are investigated.

2 Experimental

Based upon the AB₅-type (LaCePr)(NiCoMnAl)₅

hydrogen alloy, (LaCePr)(NiCoMnAl)_x ($x=4.8, 4.9, 5.0, 5.1, 5.2$) non-stoichiometric alloys were designed and prepared by induction melting under argon shield of pure elements in a water-cooled copper crucible. The ingots were melted and turned over three times for homogeneity. Then the ingots were crushed and mechanically grounded to a particle size of less than 74 μm .

The crystal structures of (LaCePr)(NiCoMnAl)_x (replaced by AB_x in the following) were analyzed by X-ray diffractogram.

2.1 Preparation of the MH electrode

0.1 g alloy powder was mixed uniformly with 0.3 g nickel powder in proportion of 1:3, and then they were pressed by 20 MPa into the negative plate 13 mm in diameter and 0.5 mm in thickness. The counter electrode was formed by the NiOOH/Ni(OH)₂, whereas the reference electrode was Hg/HgO electrode filled with 6 mol/L KOH solution.

2.2 Electrochemical measurements

After activation at room temperature (20 $^{\circ}\text{C}$), the electrodes were charged for 7 h at 60 mA/g (0.2C), then some were kept for 30 min at room temperature and the others were placed in the low temperature cabinets for 4 h and 6 h at the temperature of $-20\text{ }^{\circ}\text{C}$ and $-40\text{ }^{\circ}\text{C}$,

respectively, then the samples were discharged at various rates (0.2C, 0.5C, 1C and 3C) to -0.6 , -0.6 , -0.5 , -0.4 voltage with respect to Hg/HgO reference electrode.

Pressure-composition-temperature(PCT) curves of the alloys were determined by electrochemical methods[6]. The thermodynamics parameters including the equilibrium pressure of the hydrogen absorption or desorption and the change of enthalpy were calculated.

The electrochemical impedance spectroscopies of the 50%DOD (depth of discharge) electrodes were measured and the scan frequency was from 100 kHz to 1 mHz.

3 Results and discussion

3.1 Discharge capacity of electrode

Fig.1 shows the variation of the discharge capacity, versus the value of x of the AB_x alloy. It can be seen that with the increase of x , the discharge capacities curves of various rates (0.2C, 0.5C, 1C, 3C) at 20 °C and -20 °C, as well as that of 0.2C at -40 °C, present the shape of character M. Over-stoichiometric alloys and under-stoichiometric alloys exhibit better high-rate discharge capability and low-temperature performance compared with that of the stoichiometric alloy ($x=5.0$). The alloy with $x=5.1$ shows the excellent electrochemical capability, of which the 3C discharge capacity reaches 284 mA·h/g at 20 °C, and 0.2C also reaches 233 mA·h/g at -40 °C.

3.2. Structural analysis

The XRD patterns of AB_x metal-hydride alloys are given in Fig.2. The results show that all alloys keep the typical single phase of $CaCu_5$. The lattice parameters and lattice volume are calculated and compiled in Table 1. It can be seen that with the increase of the value of x , The lattice parameters c first increase and then decrease, which is similar to the behaviour of the low-temperature discharge capacity. On the contrary, the lattice parameters a exhibit the opposite phenomenon. The variation of lattice parameters and volume as a function of x is showed in Fig.3. The alloys with smaller lattice parameter a and larger lattice parameter c show better low-temperature performances and high-rate discharge capabilities.

Table 1 Smaller (a) and bigger (b) lattice parameters and lattice volume(V) of AB_x metal- hydride alloys

Compound	a/nm	c/nm	V/nm^3
$AB_{4.8}$	0.499 5	0.404 6	0.087 43
$AB_{4.9}$	0.499 3	0.405 8	0.087 60
$AB_{5.0}$	0.502 6	0.415 2	0.087 83
$AB_{5.1}$	0.499 9	0.403 9	0.087 40
$AB_{5.2}$	0.500 0	0.402 3	0.087 10

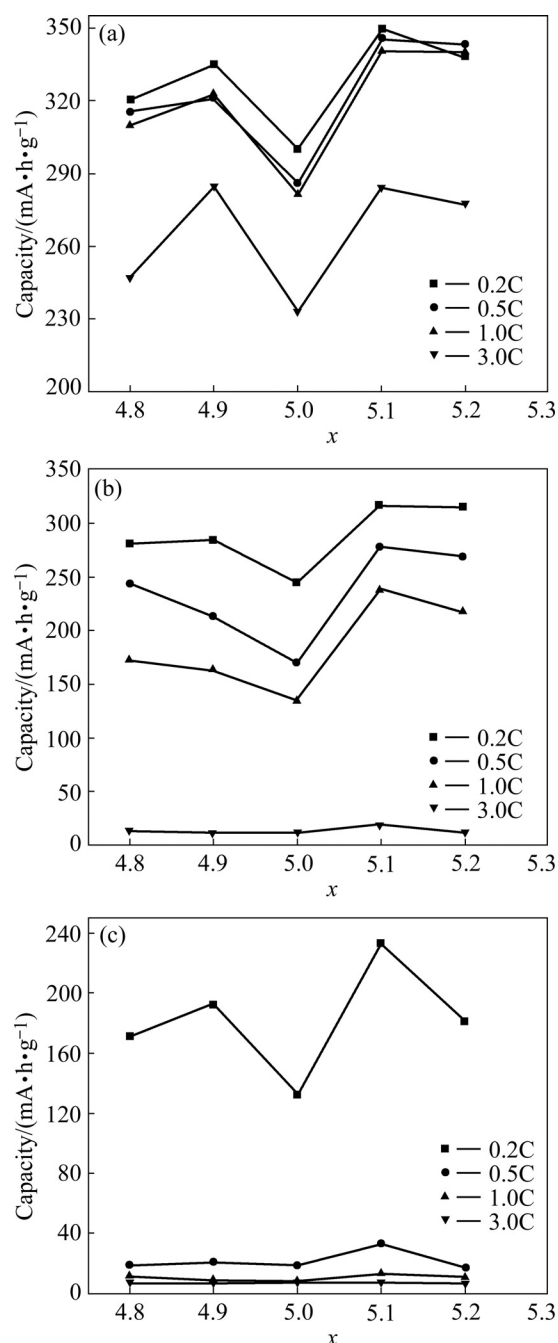


Fig.1 Variation of different rate discharge capacities with x for AB_x alloys at different temperatures: (a) 20 °C; (b) -20 °C; (c) -40 °C

3.3 Thermodynamic properties

The pressure-composition-temperature(P—C—T) curves measured at 20, -20 and -40 °C are shown in Fig.4.

The equilibrium pressure of hydrogen absorption or desorption (P_{eq}) are defined as following equation[7]:

$$P_{eq} = \frac{P_2 + P_1}{2} \quad (1)$$

where P_2 and P_1 are the pressures of inflection points of curves in Fig.4. The variation of P_{eq} versus the value

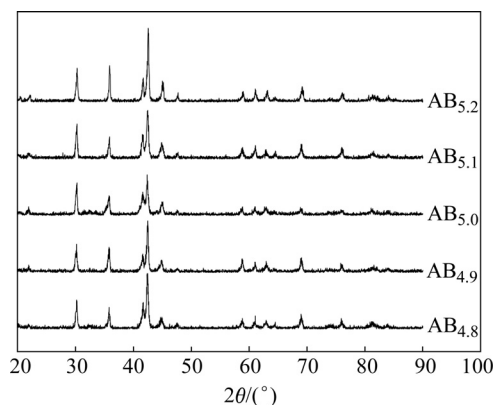


Fig.2 XRD patterns of of AB_x metal-hydride alloys

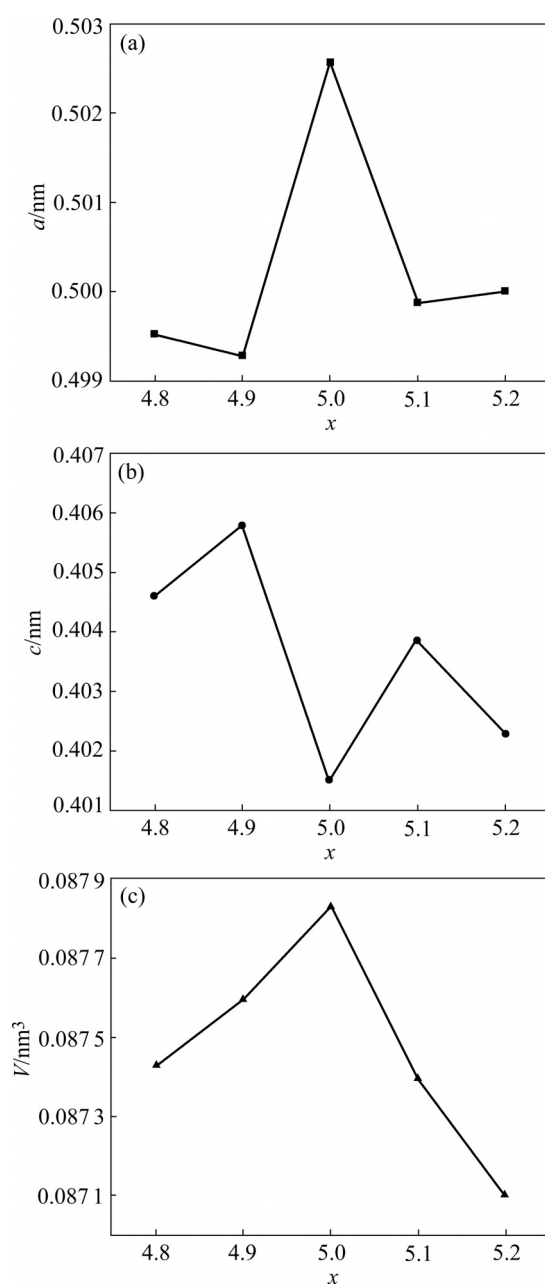


Fig.3 Variation of lattice parameters (a , c) and lattice volume (V) of AB_x metal-hydride alloys as function of x : (a) a ; (b) c ; (c) V

of x is given in Fig.5. According to P_{eq} , Van't Hoff curves of AB_x alloys is presented in Fig.6, and the variation of the change of enthalpy (ΔH) of AB_x alloys as a function of x is given in Fig.7.

The results show that over-stoichiometric alloys and under-stoichiometric alloys present higher equilibrium pressure (P_{eq}) and less absolute value of the change of enthalpy ($|\Delta H|$). The alloy with $x=5.0$ exhibit the lowest P_{eq} and the highest $|\Delta H|$, while the alloy with $x=5.1$ shows the higher P_{eq} and the least $|\Delta H|$, and its $|\Delta H|$ reaches 28.9 kJ/mol. With the increase of the value of x , the change of enthalpy (ΔH) shows the same tendency as that of the discharge capacities, which indicates that there is a close relationship between the electrochemical property and the thermodynamic performance for non-stoichiometric alloys. The alloys with higher equilibrium pressure and less absolute value of enthalpy display higher high-rate discharge capacities and low-temperature capabilities, which may be ascribed to the poor stability of metal hydrides materials.

3.4 Electrochemical impedance spectroscopy

Electrochemical impedance is a powerful tool for characterization of metal-hydride electrodes, and has been widely used[8–10]. According to the mathematical model developed by WANG et al[11], the charge-transfer resistance, R_t , could be calculated from the electrochemical impedance spectroscopy(EIS) by constructing the equivalent circuit, and the exchange current density (I_0) could be defined as following equation[12]:

$$I_0 = \frac{RT}{mFR_t} \quad (2)$$

where R , T , m and F are the gas constant, the absolute temperature, the effective mass of material and the Faraday constant, respectively.

Fig.8 shows the Nyquist plot of AB_x metal-hydride alloys with 50%DOD, and the calculated R_t and I_0 are listed in Table 2 and Table 3.

Fig.9 shows the variation of the exchange current density (I_0) of AB_x metal-hydride alloys (50%DOD) as a function of x . With the increase of x , the exchange current densities of AB_x alloys present the same tendency as that of the discharge capacities, and the exchange current densities of the non-stoichiometric alloys are higher than that of AB_5 alloy. The high-rate discharge capacities increase gradually with the increase of the exchange current density[13–15]. It is well known that the high exchange current density indicates not only the high reaction rate of the electrode, i.e. high rate charge-discharge capability, but also a low degradation rate of electrode performance[16]. Thus, we can conclude that the non-stoichiometry is beneficial to the

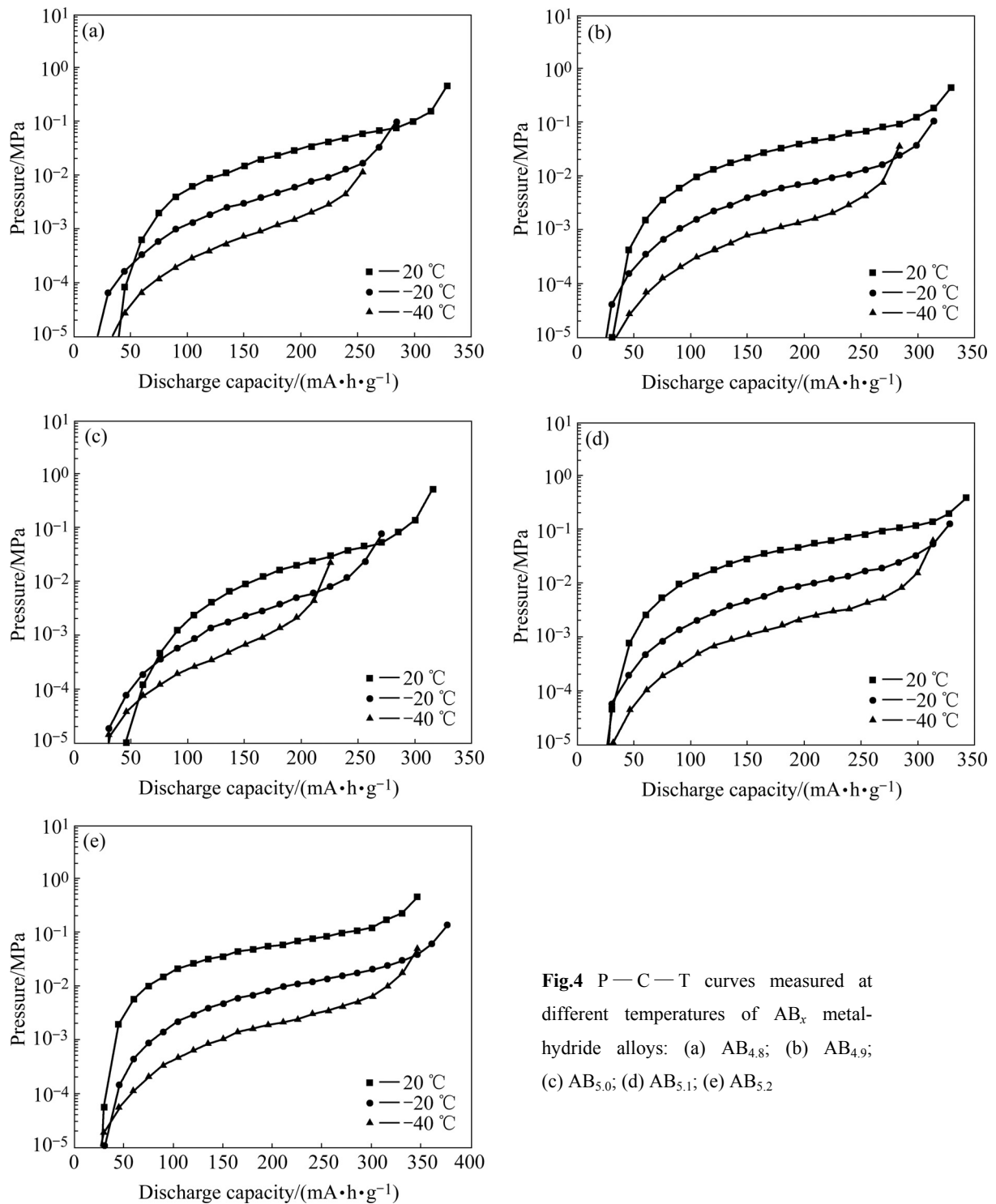


Fig.4 P—C—T curves measured at different temperatures of AB_x metal-hydride alloys: (a) AB_{4.8}; (b) AB_{4.9}; (c) AB_{5.0}; (d) AB_{5.1}; (e) AB_{5.2}

Table 2 Charge-transfer resistance (R_t) of AB_x metal-hydride alloys (50%DOD) at different temperatures

T/K	R_t/Ω				
	AB _{4.8}	AB _{4.9}	AB _{5.0}	AB _{5.1}	AB _{5.2}
293	0.276 6	0.246 6	0.405 4	0.267 9	0.477 9
253	3.531 0	3.332 0	7.944 0	2.966 0	5.067 0
233	16.810 0	13.100 0	41.440 0	11.670 0	21.520 0

Table 3 Exchange current density (I_0) of AB_x metal-hydride alloys (50%DOD) at different temperatures

T/K	$I_0/(\text{mA}\cdot\text{g})$				
	AB _{4.8}	AB _{4.9}	AB _{5.0}	AB _{5.1}	AB _{5.2}
293	913.2	1024.3	623.1	942.9	528.6
253	59.4	65.5	27.5	73.5	43.1
233	11.9	15.3	4.84	17.2	9.3

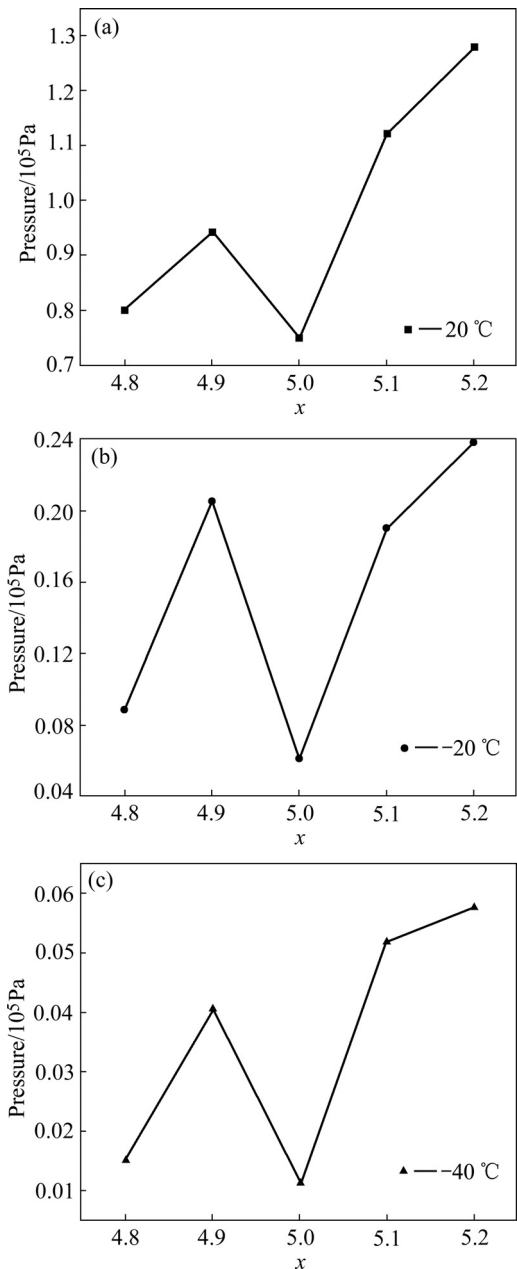


Fig.5 Variation of equilibrium pressures of AB_x metal-hydride alloys as function of x : (a) 20°C ; (b) -20°C ; (c) -40°C

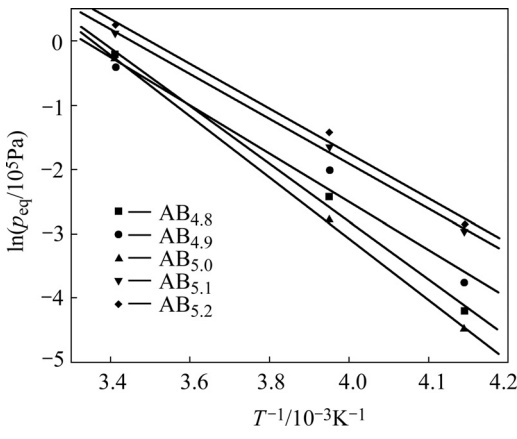


Fig.6 Van't Hoff curves of AB_x metal-hydride alloys

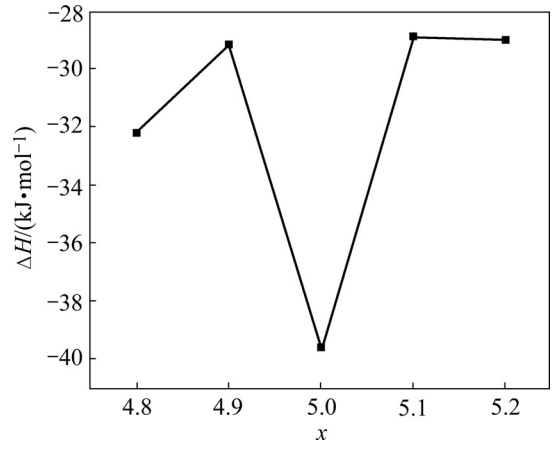


Fig.7 Variation of enthalpy (ΔH) of AB_x metal-hydride alloys as function of x

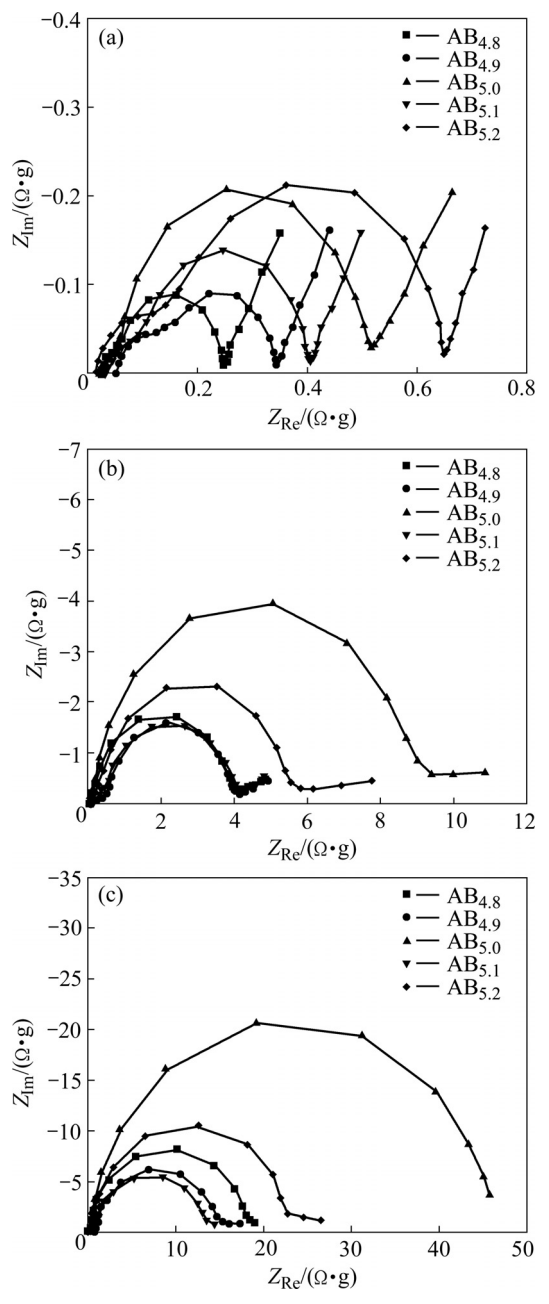


Fig.8 Nyquist plots of AB_x metal-hydride alloys (50%DOD)

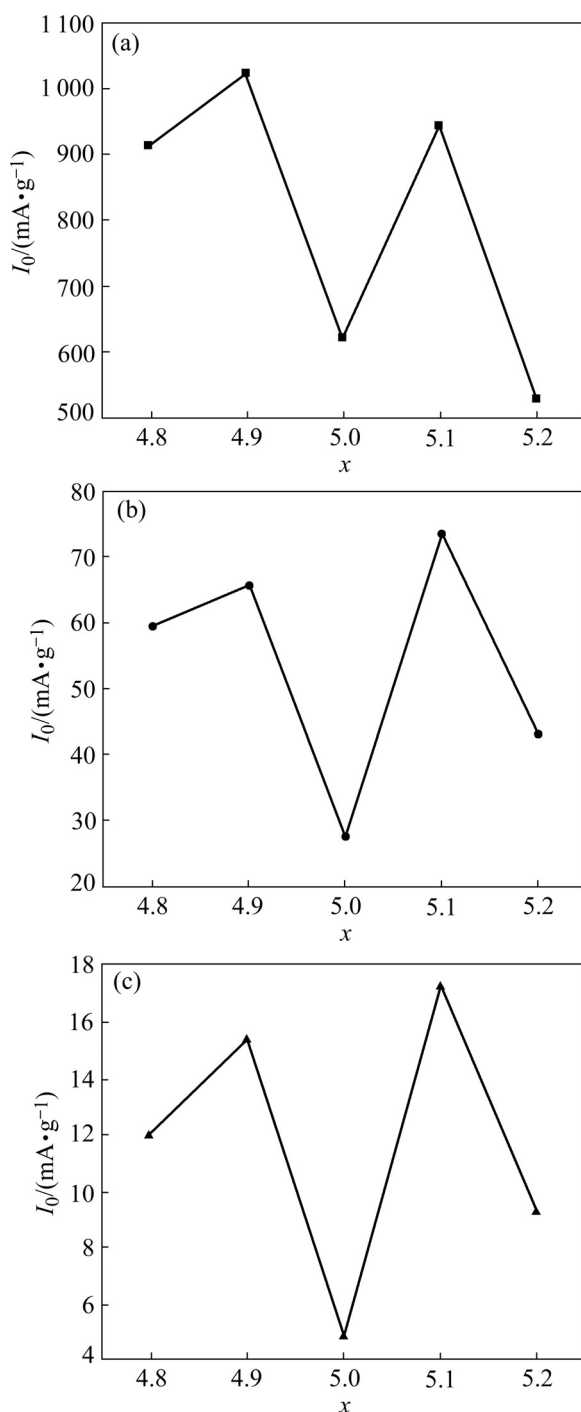


Fig.9 Variation of exchange current density (I_0) of AB_x metal-hydride alloys as function of x : (a) 20 °C; (b) -20 °C; (c) -40 °C

increase of electrodes exchange current density, which can improve the low-temperature performance and high-rate discharge ability.

4 Conclusions

1) With the increase of x , the discharge capacities of AB_x electrodes present the shape of “M”. Under-

stoichiometric and over-stoichiometric alloys present the better low-temperature performance and the higher high-rate discharge ability compared with that of the stoichiometric alloy ($x=5.0$).

2) Non-stoichiometric alloys possess smaller lattice parameter a and bigger lattice parameter c , which may be beneficial to their low-temperature performance and high-rate discharge capability.

3) With the increase of x , the change of enthalpy (ΔH) and the exchange current density (I_0) of AB_x alloys show the same tendency as that of the discharge capacities, which indicate that there exists a close relationship between the ΔH , I_0 , low-temperature performance and high-rate discharge ability.

4) The $\text{AB}_{5.1}$ alloy, with the least $|\Delta H|$ and highest I_0 , shows the best discharge capability. Its 3C discharge capacity reaches 284 $\text{mA} \cdot \text{h/g}$ at 20 °C, while 0.2C reaches 233 $\text{mA} \cdot \text{h/g}$ at -40 °C.

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